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Spectral shape analysis of ultraviolet luminescence in *n*-type ZnO:Ga

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Thin films of laser molecular-beam epitaxy grown *n*-type Ga-doped ZnO were investigated with respect to their optical properties. Intense room-temperature photoluminescence (PL) in the near-band edge (NBE) region was observed. Moreover, its broadening of PL band was significantly larger than predicted by theoretical results modeled in terms of potential fluctuations caused by the random distribution of donor impurities. In addition, the line shape was rather asymmetrical. To explain these features of the NBE bands, a vibronic model was developed accounting for contributions from a series of phonon replicas. © 2005 American Institute of Physics.

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Recently, optical properties of near-band-edge (NBE) recombination in ZnO epitaxial layers have been extensively investigated. Compared to undoped ZnO, the properties of heavily donor doped ZnO, such as phonon interactions, are not well understood. In our previous work, we reported the observation of intense NBE photoluminescence (PL) for ZnO:Ga at room temperature (RT). Figure 1(a) shows their PL spectra at four different doping levels (dashed lines). The experimental procedures¹ were identical to those adopted in our previous study and their details were reported elsewhere.² The shape of the PL band is asymmetrical. The linewidths of the PL increase from 154 to 195 meV with an increase in Ga concentration (n_{Ga}). Previously, we estimated the “theoretical” broadening for the PL in terms of potential fluctuations caused by the random distribution of donor impurities (cf. Fig. 3 of Ref. 1). These values are also listed in Table I. However, the calculated widths (σ_1) were smaller than the experimentally observed broadening.

These theoretical widths might be exceedingly underestimated because the model might be too much simplified. We here adopt another approach properly taking the microscopic fluctuation of donor concentration into account; the thermalization redistribution model developed by Zimmermann *et al.*³ According to this model, there is a relationship between the Stokes shift and the full-width at half maximum (FWHM) of PL. One can estimate the energy scale of the band potential profile fluctuation from the Stokes shift and predict consistently the FWHM. Figure 2 shows a Stokes shift of the luminescence plotted against the gallium concentration. Obviously, the shift energy increases with incorporation of the donor dopants.

At sufficiently high temperatures, the above-mentioned relationship is indicated as $E_{\text{Stokes}} = -\sigma^2/kT$. The nomenclatures (k and T) take their conventional meanings. For example, the Stokes shift for the sample with the n_{Ga} of $8.0 \times 10^{18} \text{ cm}^{-3}$ is 22 meV, leading to σ of 24 meV. It also yields the temperature independent value of $2\sigma\sqrt{\ln 4} \approx 46 \text{ meV}$ in the FWHM. This can be also supported by the three-dimensional Monte Carlo simulation. We now simulate a PL spectrum.

Miller-Abraham’s rate for phonon-assisted exciton tunneling between the initial and final states i and j with the energies of E_i and E_j was adopted to simulate the hopping events of excitons or of photocreated carriers:

$$\nu_{ij} = \nu_0 \exp\left(-\frac{2r_{ij}}{\alpha} - \frac{(\epsilon_i - \epsilon_j + |\epsilon_i - \epsilon_j|)}{2kT}\right). \quad (1)$$

Here r_{ij} is the distance between the localized states, α is the decay length of the wave function, and ν_0 is the attempt-escape frequency. Hopping was simulated over a randomly generated set of localized states with the density of N . Dispersion of the localization energies was assumed to be in accordance with a Gaussian distribution,

$$g(\epsilon) \propto \exp(-(\epsilon - E_0)^2/2\sigma^2). \quad (2)$$

With the peak positioned at the mean excitonic energy E_0 and the dispersion parameter (the energy scale of the band potential profile fluctuation) σ . For each generated exciton, the hopping process terminates by recombination with the probability τ_0^{-1} and the energy of the localized state, where the recombination has taken place is scored to the emission spectrum.

In addition, the thermal broadening effect was implemented by convoluting the above spectrum with a Gaussian curve, which describes the thermal part with a width of $1.8 kT$.⁴ Because the above equation contains the term of thermal activation, at a glance, it seems that it is unnecessary

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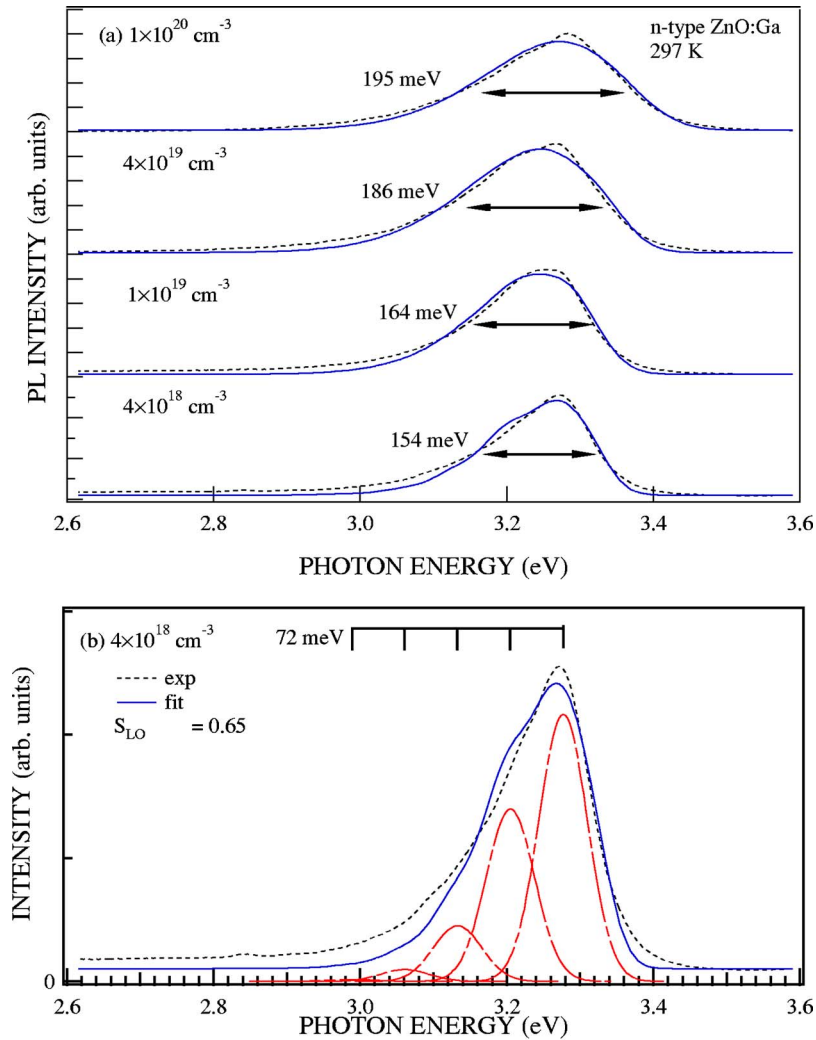


FIG. 1. (Color online) (a) Room-temperature photoluminescence spectra (dashed curves) of *n*-type ZnO doped with different Ga concentrations. Also shown are the results of the fit to the data using a “vibronic” model (solid lines). (b) The lowermost PL curve of frame (a) with individual contribution of the emission lines (dash-dotted lines).

to perform the convolution. However, because Kazlauskas *et al.* adopted this treatment, we follow it in this work.⁵ Meanwhile, the overall FWHM of the PL band is given:

$$\text{FWHM}(\sigma_2) \approx [(2\sigma\sqrt{\ln 4})^2 + (1.8 kT)^2]^{1/2}. \quad (3)$$

The solid line in Fig. 3 represents the simulated result obtained for the following values of parameters: $N\alpha^3=0.1$, $\tau_0\nu_0=10^4$, and $\sigma\approx 24$ meV. The line shape of resulting spectrum was independent of the choices of the $N\alpha^3$ and $\tau_0\nu_0$ parameters. The simulated curve could be well approximated by the Gaussian function, which is not surprising if considering the measurement temperature (297 K). We summarize

TABLE I. Sample specification and main characteristics for the four *n*-ZnO samples. Ga concentration (n_{Ga}), theoretical broadening σ_1 , broadening evaluated from the Monte Carlo simulation σ_2 , zero-phonon peak energy E , and the deduced Huang-Rhys factor S_{LO} .

n_{Ga} (cm^{-3})	σ_1 (meV)	σ_2 (meV)	E (eV)	S_{LO}
8.0×10^{18}	77.7	73.5	3.277	0.64
2.0×10^{19}	97.1	115	3.272	0.78
8.0×10^{19}	150	139	3.297	1.1
1.5×10^{20}	188	170	3.345	1.4

the FWHMs (σ_2) of the simulated PL spectra for the samples at four different doping levels in Table I.

The simulated broadening (σ_2) is still smaller than that of experiments. This discrepancy is thought to be due to the contribution of the phonon replicas. In addition, the simulated line shape is rather symmetrical, which is in great contrast to the asymmetrical shape observed in the experimental

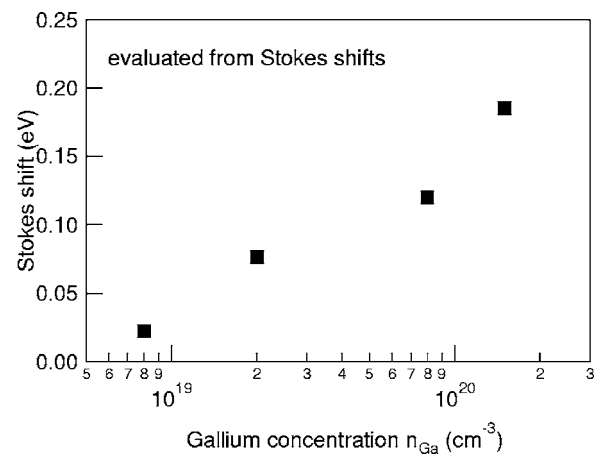


FIG. 2. The PL Stokes shifts (closed squares) plotted against the Ga concentration.

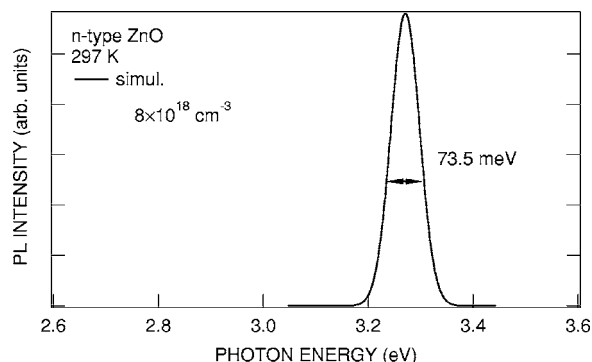


FIG. 3. Simulated PL spectrum under stationary excitation. The Ga concentration is shown in the figure.

spectra. In compensated semiconductors, even a low-temperature emission spectrum shows very broad feature. Its line shape is sometimes asymmetrical. Kuskovsky and his co-workers⁶ have submitted a theory reproducing such an asymmetrical line shape. On the other hand, our experiments were performed at room temperature. Therefore, it is natural to think that the shape of an unresolved zero-phonon band is rather symmetrical because of the thermalization effects. We did not estimate the width and the asymmetry of our PL using their theory because, unfortunately, it is applicable only to the case of extremely low temperatures and because our samples are not very severely compensated.¹ We think that the zero-phonon peak and its phonon replicas are not spectrally resolved well in the spectrum because intensity of a one-phonon replica is comparable to that of a zero-phonon band, leading to the larger experimental broadening.

The typical consequence of the interaction with phonons is the appearance of phonon-assisted emissions in a PL spectrum. We infer that the coupling with longitudinal-optical (LO) phonon is very efficient in this highly polar material. The reported coupling constant of ZnO is nearly four times stronger than that of GaN and about eight times stronger than that of ZnSe.⁷

For simplicity, we took only the terms of longitudinal optical (LO) phonons into account,^{8–10} whose interaction is the strongest. In this case, the transition energy of the replicas will be $\hbar\omega = E_{\text{ZPL}} - \eta_{\text{LO}}E_{\text{LO}}$, where $E_{\text{LO}} = 72$ meV in ZnO, E_{ZPL} is the energy of the zero-phonon peak and η_{LO} is an integer.

Each individual emission line in the spectrum was modeled by a Gaussian function with linewidth σ_1 (cf. Table I), which can be justified by the Gaussian-type line shape observed in our simulated spectrum. The values of σ_1 were determined from the above-mentioned theory because these values are larger than those of the Monte Carlo simulation, σ_2 . The probability of a given phonon emission is proportional to $(S_{\text{LO}}^{\eta_{\text{LO}}}/\eta_{\text{LO}}!)$, where S_{LO} is the Huang-Rhys factor for the LO phonons and η_{LO} the number of phonons emitted in a transition. The solid lines in Figs. 1(a) and 1(b) were obtained by summing all of the emission lines using an appropriate set of parameters. The donor-impurity concentrations (n_{Ga}), zero-phonon peak energies (E), and S -factors are compiled in Table I. Five dash-dotted curves in Fig. 1(b) denote the respective contributions whose peak positions are

equidistant by the E_{LO} of 72 meV. The important conclusion from this fit is that the asymmetrical and broad PL band can be explained by the contribution of LO phonon replicas. It should be noted that the Huang-Rhys factor (S_{LO}) depends on the ionic nature of semiconductors and on the separation of Fourier transformed charge distributions. This sometimes deviates from the bulk value.^{8–10} Very recently, Shan *et al.*¹¹ reported that the RT PL spectrum in undoped ZnO is dominated by the phonon replicas of the free exciton transition with the maximum at the first LO phonon replica. This corresponds to the S -factor exceeding unity. In addition, in the undoped ZnO, the zero-phonon emission becomes weaker than 1 LO-phonon-assisted annihilation processes at elevated sample temperatures.

Even if the values of σ_1 are comparable with the optical phonon energy E_{LO} , due to the high measurement temperature, we believe it is probable that the radiative transition would go via phonon-assisted path while the impurity levels are available at the same energies.

We see the shoulders at the 1 LO phonon replica position in both the fitting and experimental results, particularly in the first two cases. The “experimental” shoulder is less prominent than that of fit. The reason of this is not clear at this moment but we speculate that this is because of our assumption that the width of the phonon replicas is same with that of the zero-phonon band. We think that the broader replicas are more realistic.

In summary, gallium was applied as a source of donor impurities in the laser molecular-beam epitaxy growth of *n*-type ZnO. We presented spectroscopic measurements to study the evolution of spectra in *n*-type ZnO along with Ga concentrations. Asymmetric and broad photoluminescence spectra were fitted to a model that takes into account phonon replicas, in which the linewidths of each individual emissions have been determined mainly from the concentration fluctuation of dopants.

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